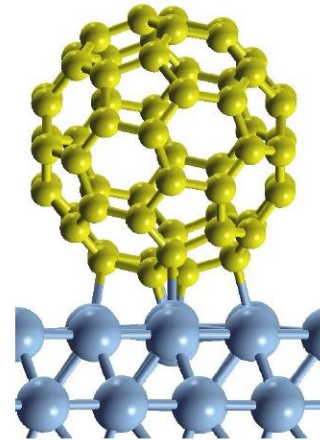


DE LA RECHERCHE À L'INDUSTRIE



MODELLING OF THE MAGNETIC ANISOTROPY OF MOLECULE/FERROMAGNETS INTERFACES



Cyrille Barreteau

Kaushik Bairagi, Vincent Repain, Amandine Bellec, Dongzhe Li, Alexander Smogunov, Ludovic Le Laurent

INTRODUCTION



ANISOTROPY IN MAGNETIC STUDIES

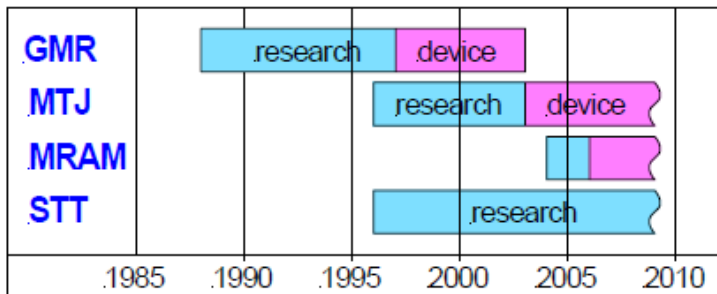
Search for permanent hard magnets (large Anisotropy) without rare earth

Replace Samarium-Cobalt, Neodyum by nanostructured « cheap » TM magnets

Spintronics

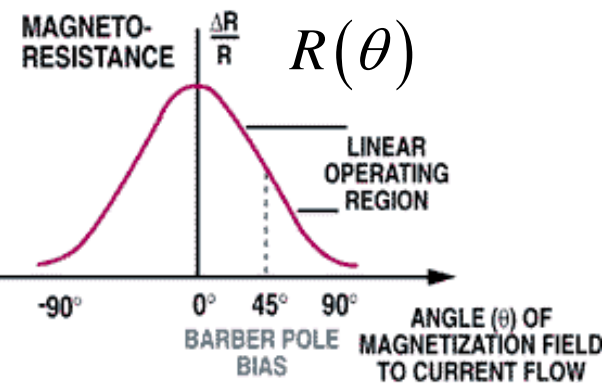
From fundamentals to applications

AMR



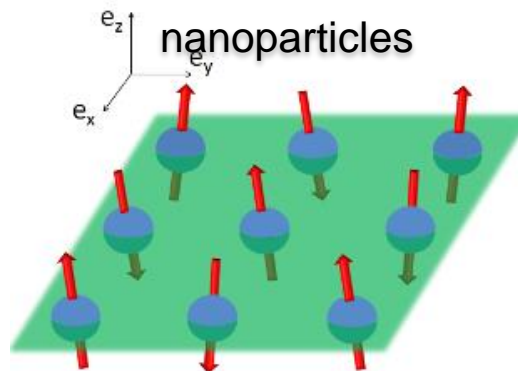
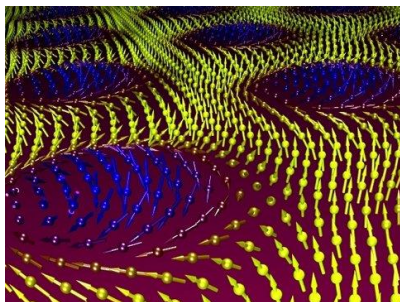
\vec{H}, \vec{M}

current I



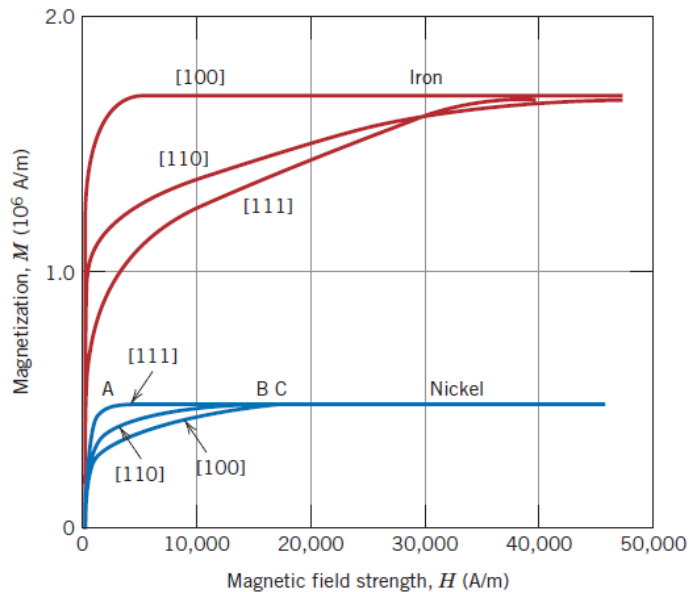
Ultimate data storage

skyrmions

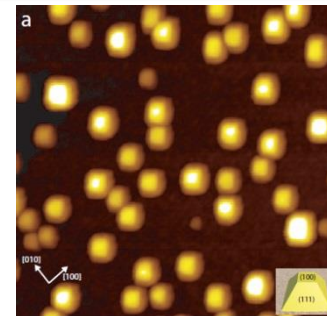
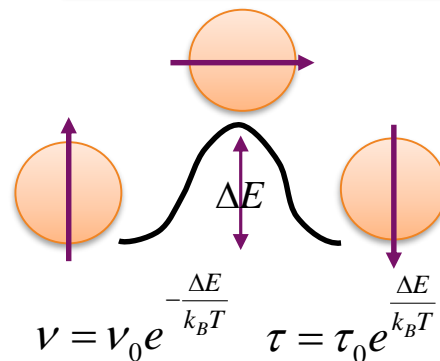


MANIFESTATION OF MAGNETIC ANISOTROPY

Magnetization curve of single crystals



Thermal stability of magnetic nanoparticles



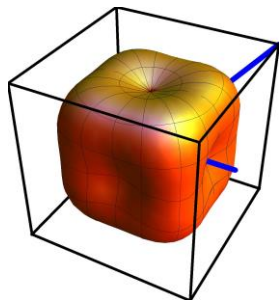
PRB 90, 205409 (2014)

τ	1s	1mn	1h	1day	1year	10years
$\frac{\Delta E}{k_B T}$	23	27	31	34	40	43

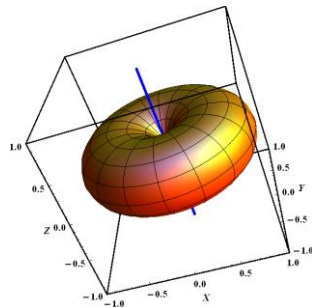
A question of energy

$$E = E(\theta, \varphi)$$

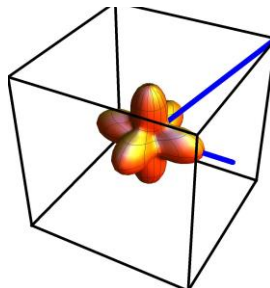
bccFe



hcpCo



fccNi



$\Delta E \sim 10^{-2} - 10^{-3} \text{ meV / atom}$ in bulk Fe, Co, Ni

Much larger in nanostructures

$\Delta E \sim \text{meV / atom}$ in bulk FePt L10

ORIGIN OF THE MAGNETIC ANISOTROPY

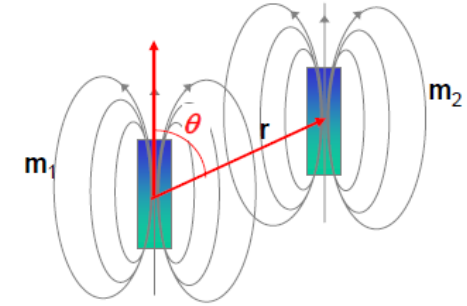
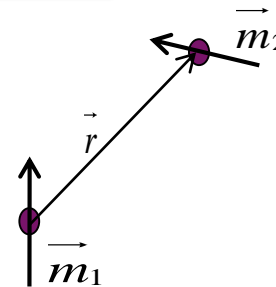
Shape anisotropy

•dipole-dipole interaction

$$E_{dip}(\vec{r}) = \frac{\mu_0}{4\pi r^3} \left[\vec{m}_1 \cdot \vec{m}_2 - \frac{3}{r^2} (\vec{m}_1 \cdot \vec{r})(\vec{m}_2 \cdot \vec{r}) \right]$$

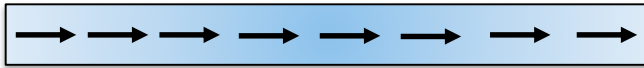
Collinear magnetism

$$E_{dip}(\theta) = \frac{\mu_0}{4\pi r^3} m_1 m_2 [1 - 3 \cos \theta]$$



For $\theta < 54.74^\circ$ FM For $\theta > 54.74^\circ$ AF

Favors elongated direction



Magnetocrystalline anisotropy

•Spin-orbit coupling

Dirac $\xrightarrow{\frac{v}{c} \ll 1}$ Schrödinger + Zeeman + Mass velocity + **Spin-orbit** + Breit

$$V_{SOC} = \underbrace{\frac{1}{2m^2c^2} \frac{1}{r} \frac{dV}{dr}}_{\xi(r)} \mathbf{L} \cdot \mathbf{S}$$

• $\xi(r)$ Very localized potential

• $\xi(r) \sim Z^4$ Increases drastically with atomic number

$$\xi_d \sim 0.05eV \text{ for } 3d \quad \xi_d \sim 0.5eV \text{ for } 5d$$

• $\mathbf{L} \cdot \mathbf{S}$ Couples spin and orbital moment
→ generates anisotropy

SHAPE ANISOTROPY VERSUS MAGNETOCRYSTALLINE ANISOTROPY

Shape anisotropy

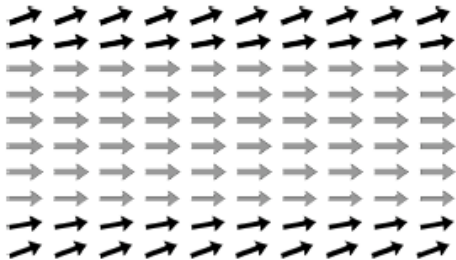
Long range interaction $\frac{1}{r^3}$

Very little dependent on crystallographic structure and local atomic environment

Very little dependent on symmetry

Very little dependent on chemical nature of atoms

Average quantity (proportional to the volume of the sample)



$$E^{\text{MCA}} \sim \text{Surface}$$

$$E^{\text{shape}} \sim \text{Volume}$$

Magnetocrystalline anisotropy

short range interaction

Strongly dependent on crystallographic structure and local atomic environment

Strongly dependent on symmetry

Strongly dependent on chemical nature of atoms

Local quantity dominated by surface and interfaces

At the end the shape anisotropy always wins!

MODEL



DFT

Plane wave basis set

Ultra-soft pseudo-potentials

Collinear and non-collinear magnetism

Spin-orbit coupling



TB

spd tight-binding

Semi-empirical model fitted on ab-initio

Collinear and non-collinear magnetism

Spin-orbit coupling



HOW TO CALCULATE MCA

Brute force method (self-consistent)

$$E^{MCA} = E_{\mathbf{n}_1}^{\text{tot}} - E_{\mathbf{n}_2}^{\text{tot}}$$

where $E_{\mathbf{n}_1}$ and $E_{\mathbf{n}_2}$ are obtained from SCF calculation including SOC

*In principle « exact » but very time consuming and hard to converge
One should use penalization techniques to obtain E_n for any direction*

Force Theorem method

$$E^{MCA} \approx \Delta E_{\text{band}} = \int^{E_F^1} E n^1(E) dE - \int^{E_F^2} E n^2(E) dE \approx \int^{E_F} (E - E_F) \Delta n(E) dE$$

Initial density is obtained from a SCF spin-collinear calculation and spin-moment further rotated to appropriate spin direction and one step SOC calculation.

Very « fast » and numerically stable but cannot be applied to systems with too large SOC.

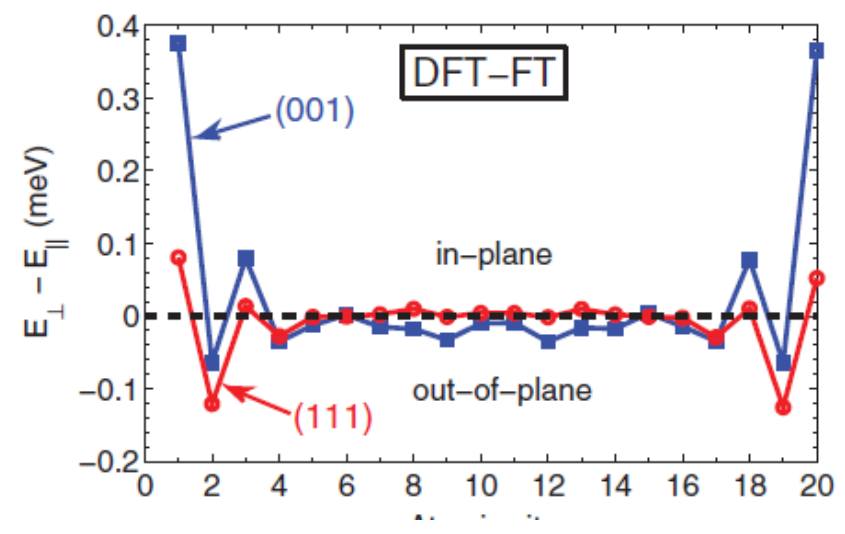
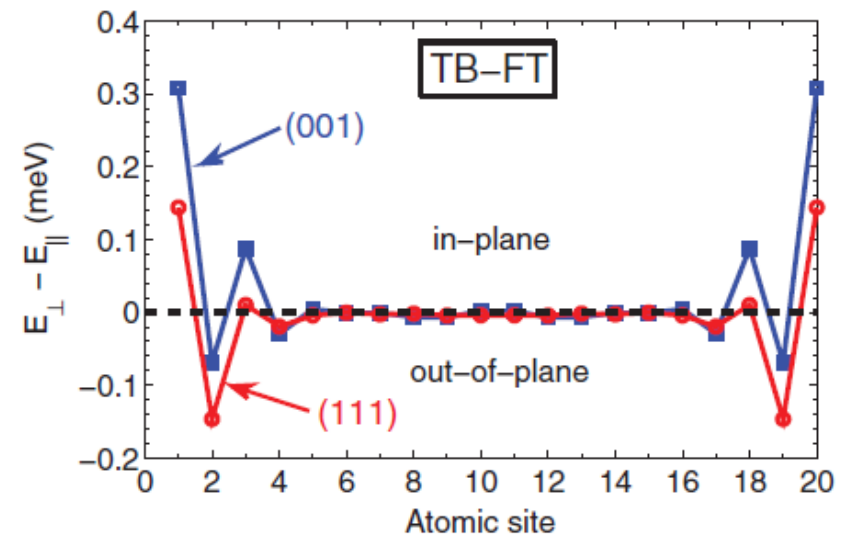
Local « site » and orbital decomposition

$$E^{MCA} \approx \sum_{i\lambda} \int^{E_F} (E - E_F) \Delta n_{i\lambda}(E) dE$$

VALIDATION OF TB MODEL

Co fcc « slabs »

TB



MCA
OF MOLECULE/FERROMAGNETS
INTERFACES



Co/Au(111) EXPERIMENT

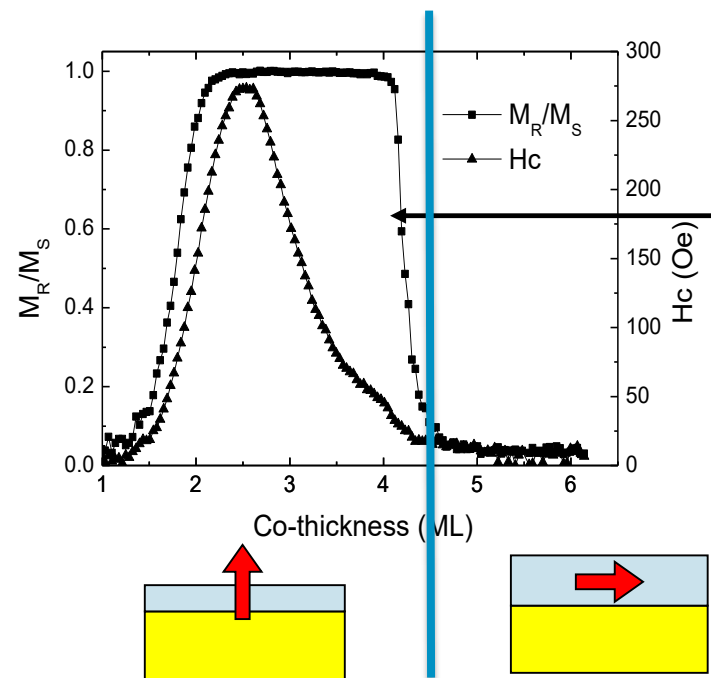
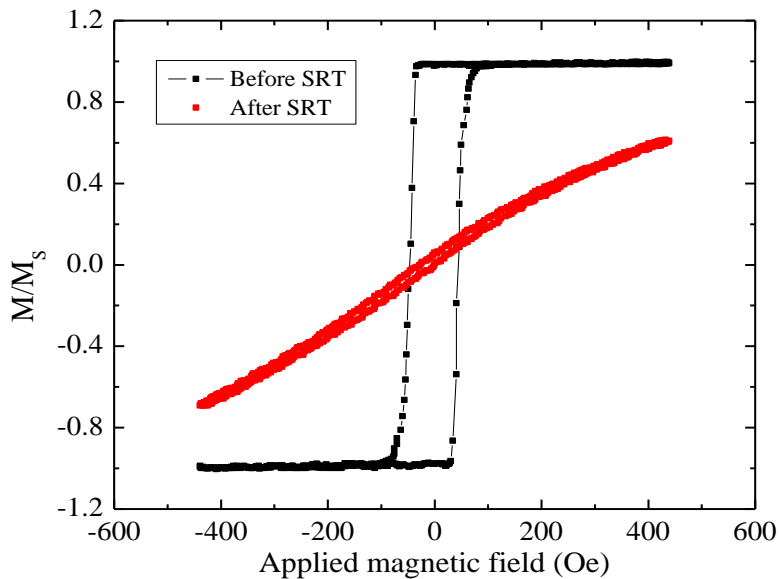
substrate Co/Au(111)

$$K^{\text{eff}} = \frac{E^{\text{MCA}} + E^{\text{shape}}}{V} = K^{\text{v}} + \frac{K^{\text{s}}}{t}$$

Co/Au(111) $K^{\text{s}} = 0.75 \text{ mJ.m}^{-2}$ and $K^{\text{v}} = -900 \text{ kJ.m}^{-3}$

Spin Reorientation Transition (SRT) at $t^* = 4.2 \text{ ML}$

Polar MOKE

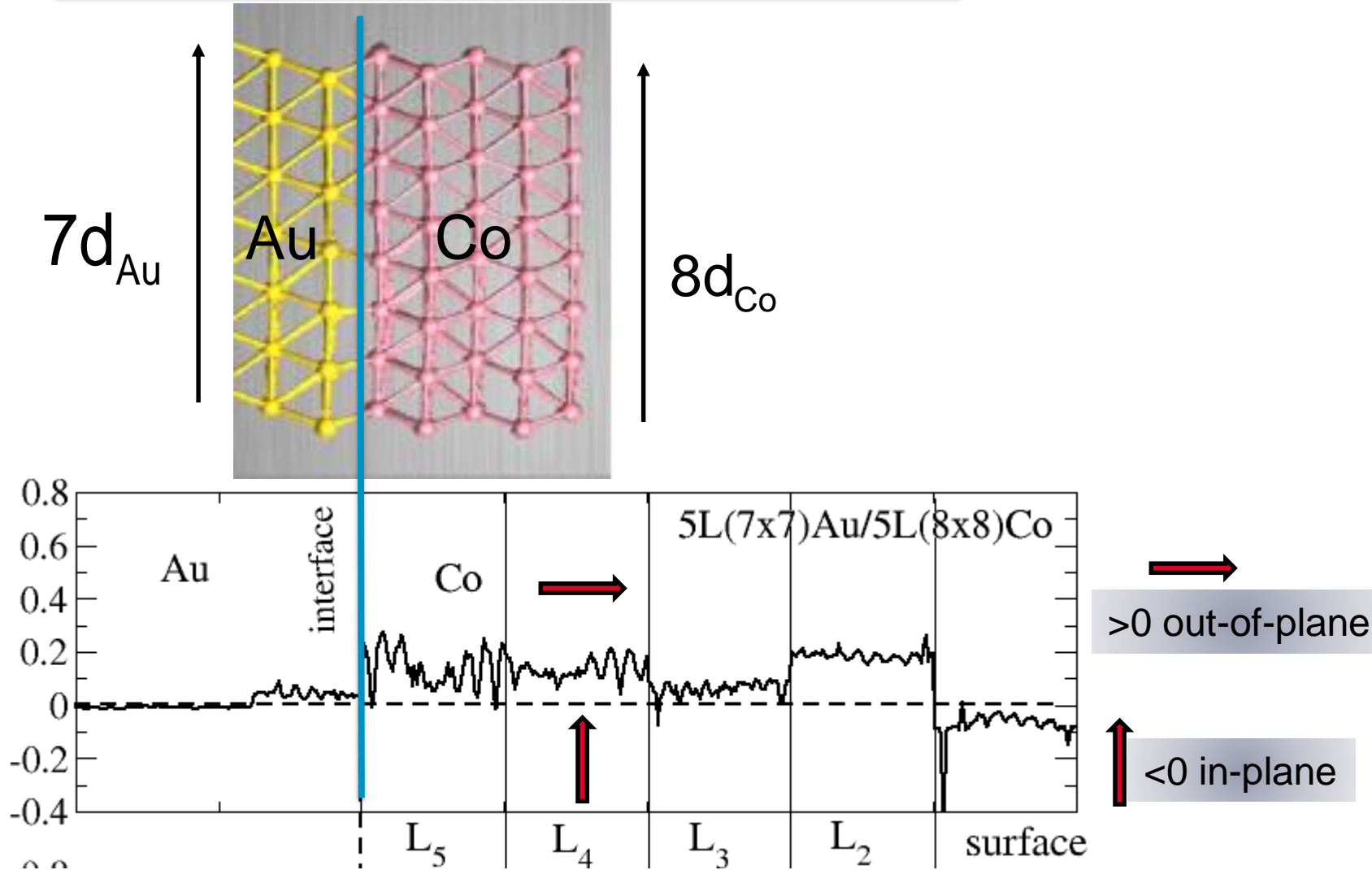


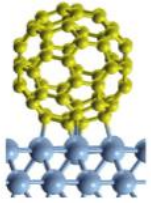
SRT

Ultra sensitive to magnetic anisotropy change close to the SRT

Co/Au(111): TB MODEL

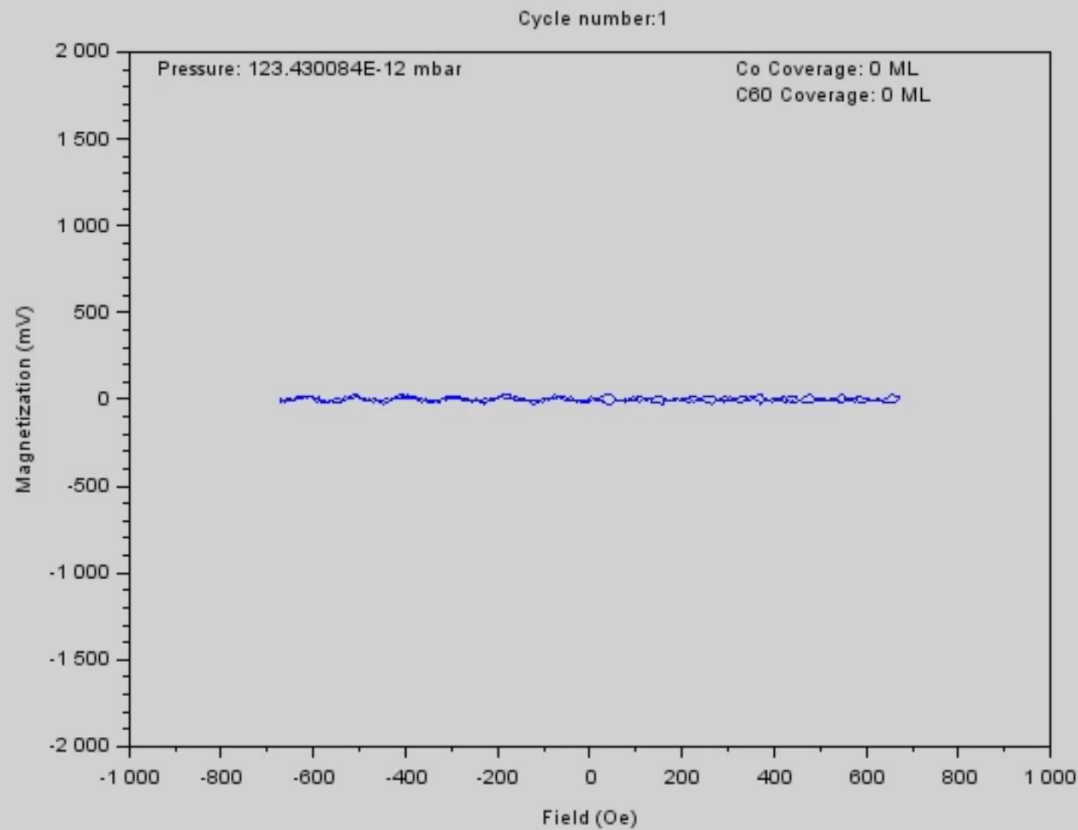
•MCA of Co(0001)/Au(111) from a realistic TB modelling



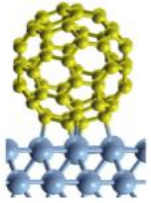


C60/Co/Au(111) EXPERIMENT

- Adsorption of C₆₀ on Co(0001)/Au(111): Experiment



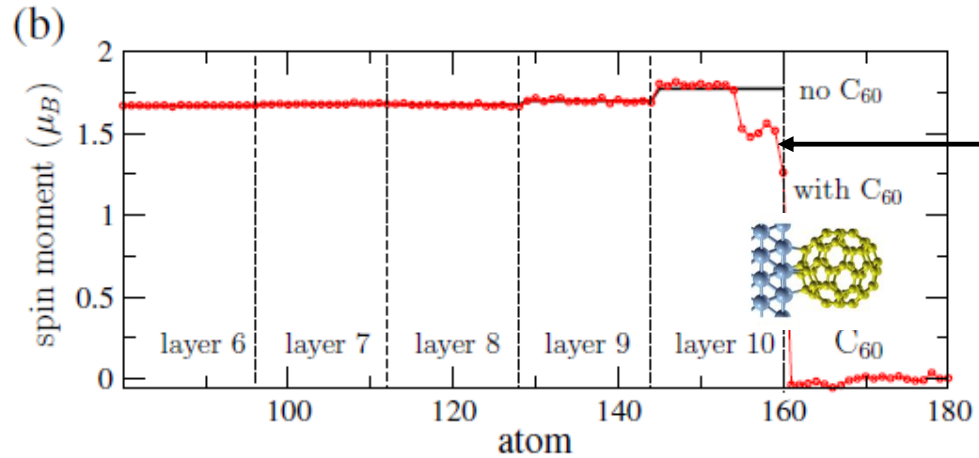
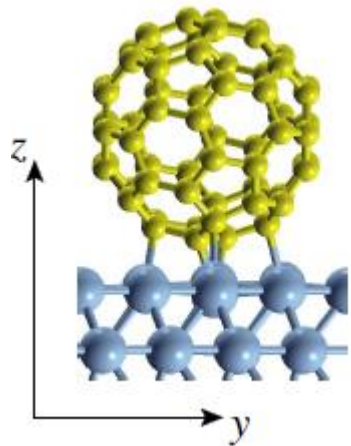
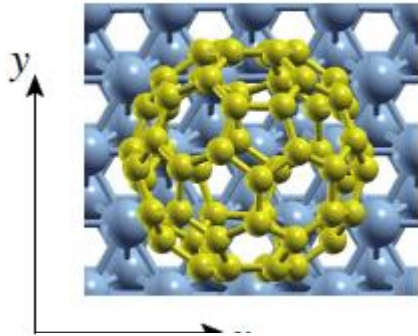
C60 favors « out-of-plane » magnetization



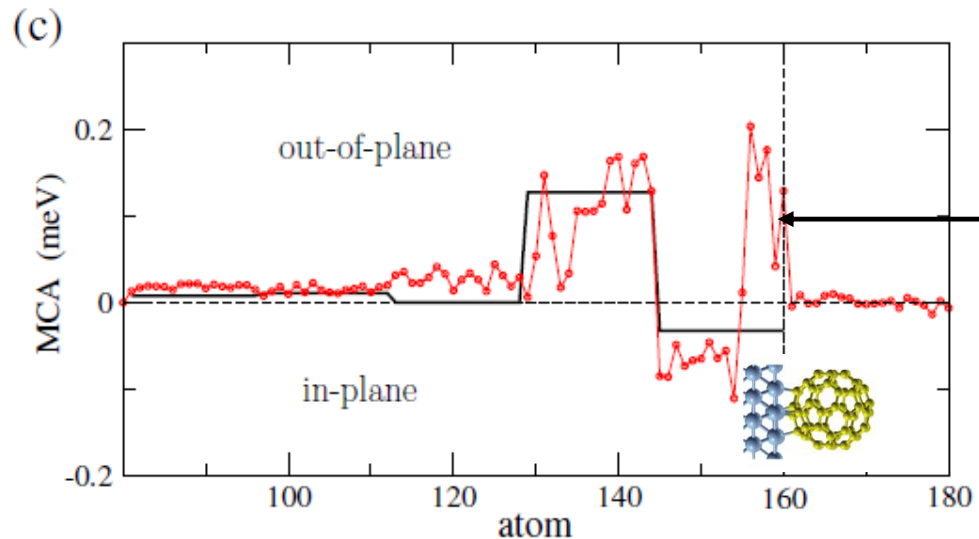
C60/Co MODELLING

• Adsorption of C₆₀ on Co(0001)/Au(111): Theory

DFT calculation:
C60/10L Co (4x4)



Decrease of magnetization

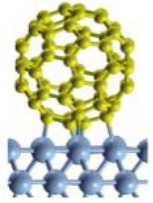


increase of "out-of-plane" MCA

$$\Delta E^{\text{ani}} = \underbrace{\Delta E^{\text{MCA}}}_{0.9} + \underbrace{\Delta E^{\text{shape}}}_{0.6} = 1.5 \text{ meV}$$

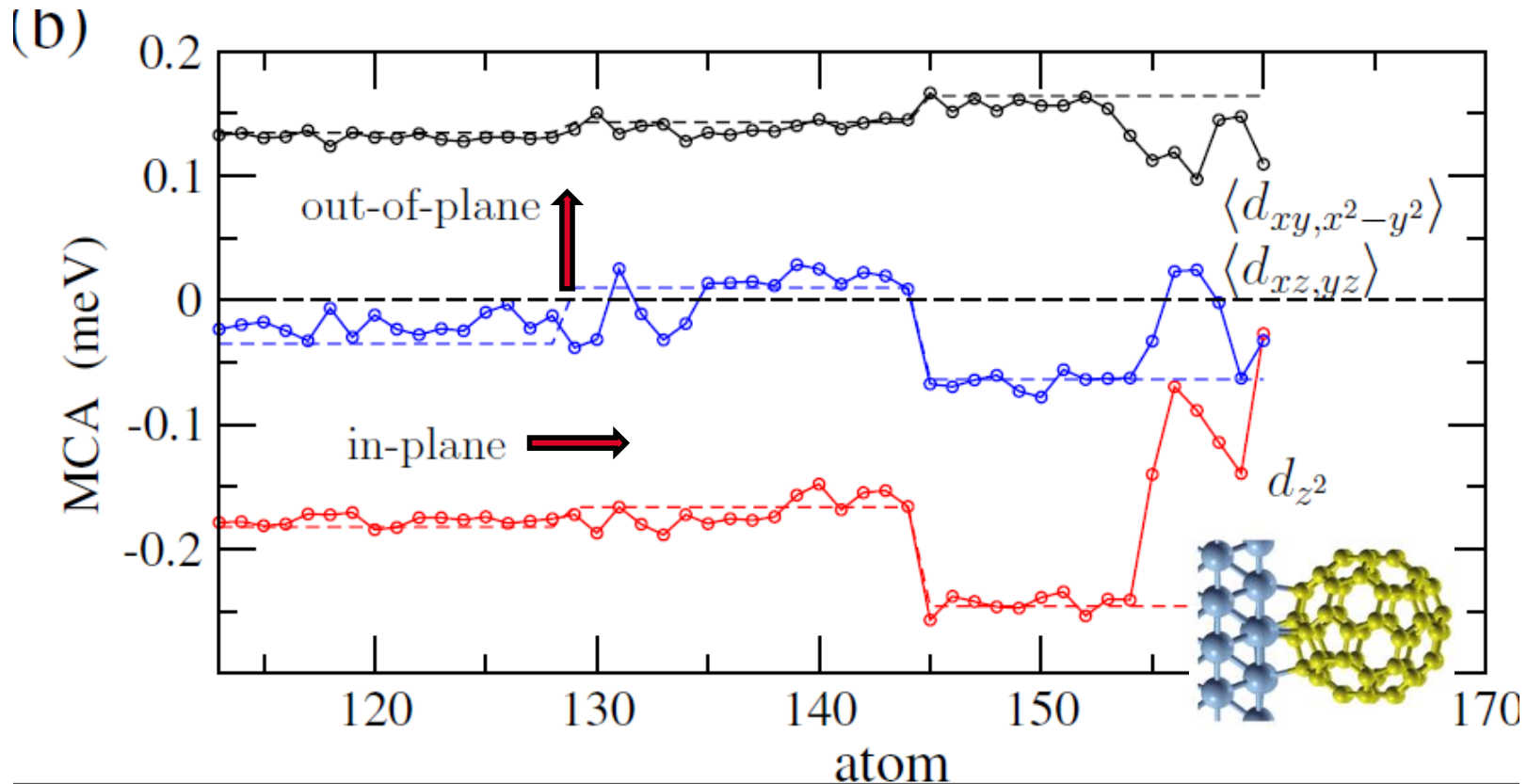
For 5x16 Co atoms

$$\Delta E^{\text{ani}} = 19 \mu\text{eV} / \text{atom}$$



C60/Co TB MODEL

Orbital-resolved MCA



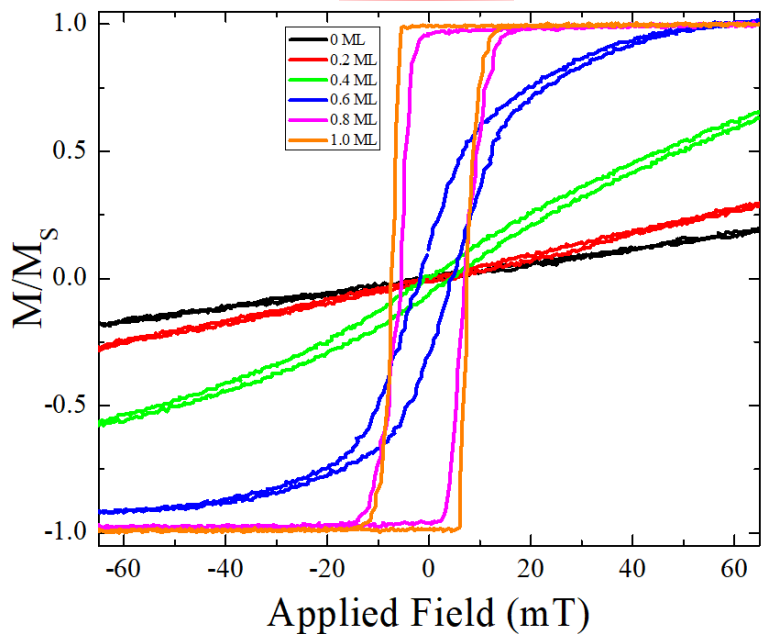
Experiment: C60 favors « out-of-plane » magnetization

Theory: C60 kills « in-plane » d_{z^2} component of MCA

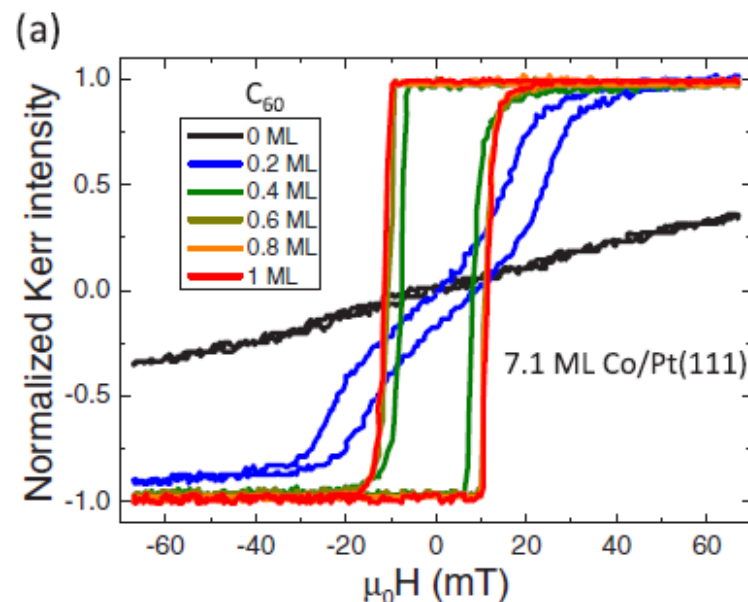
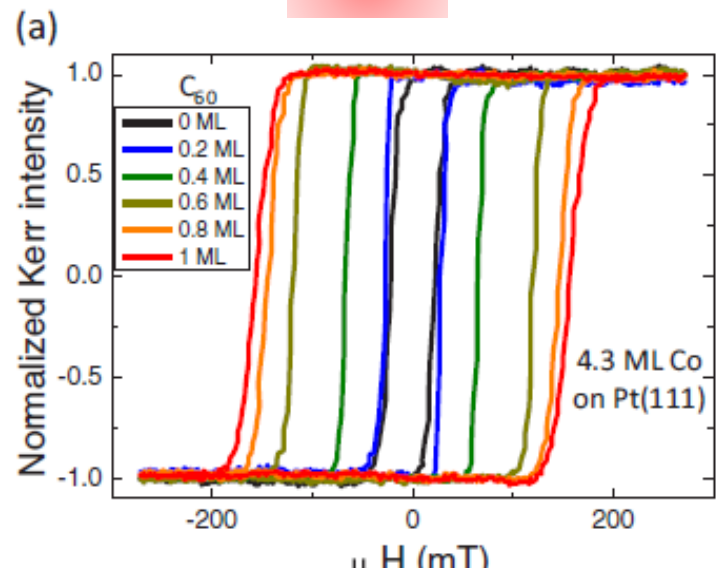
PRL 114, 247203 (2015)

$C_{60}/Co/Au(111)$ VERSUS $C_{60}/Co/Pt(111)$ EXPERIMENT

Co/Au



Co/Pt



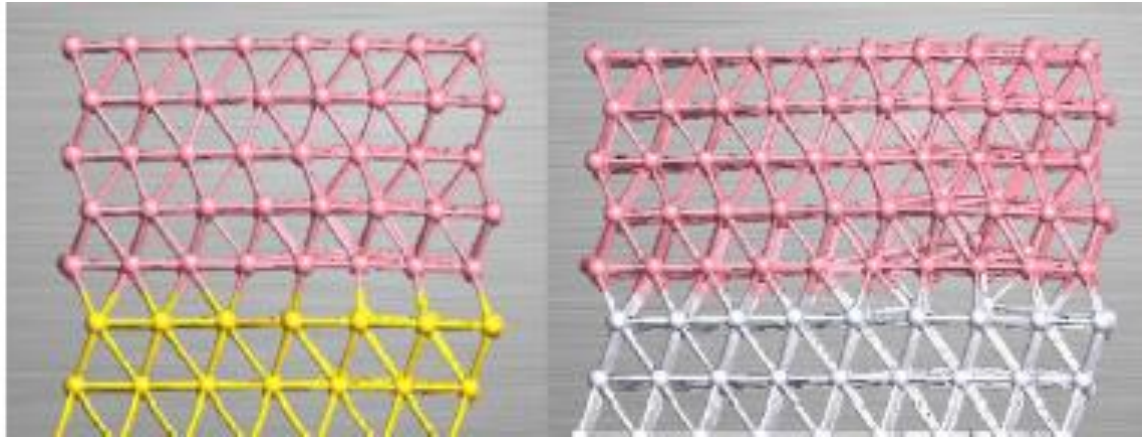
Increase of coercive field

**100% for $C_{60}/Co/Au$
700% for $C_{60}/Co/Pt$**

$C_{60}/Co/Au(111)$ VERSUS $C_{60}/Co/Pt(111)$ MODEL

Co/Au(111)

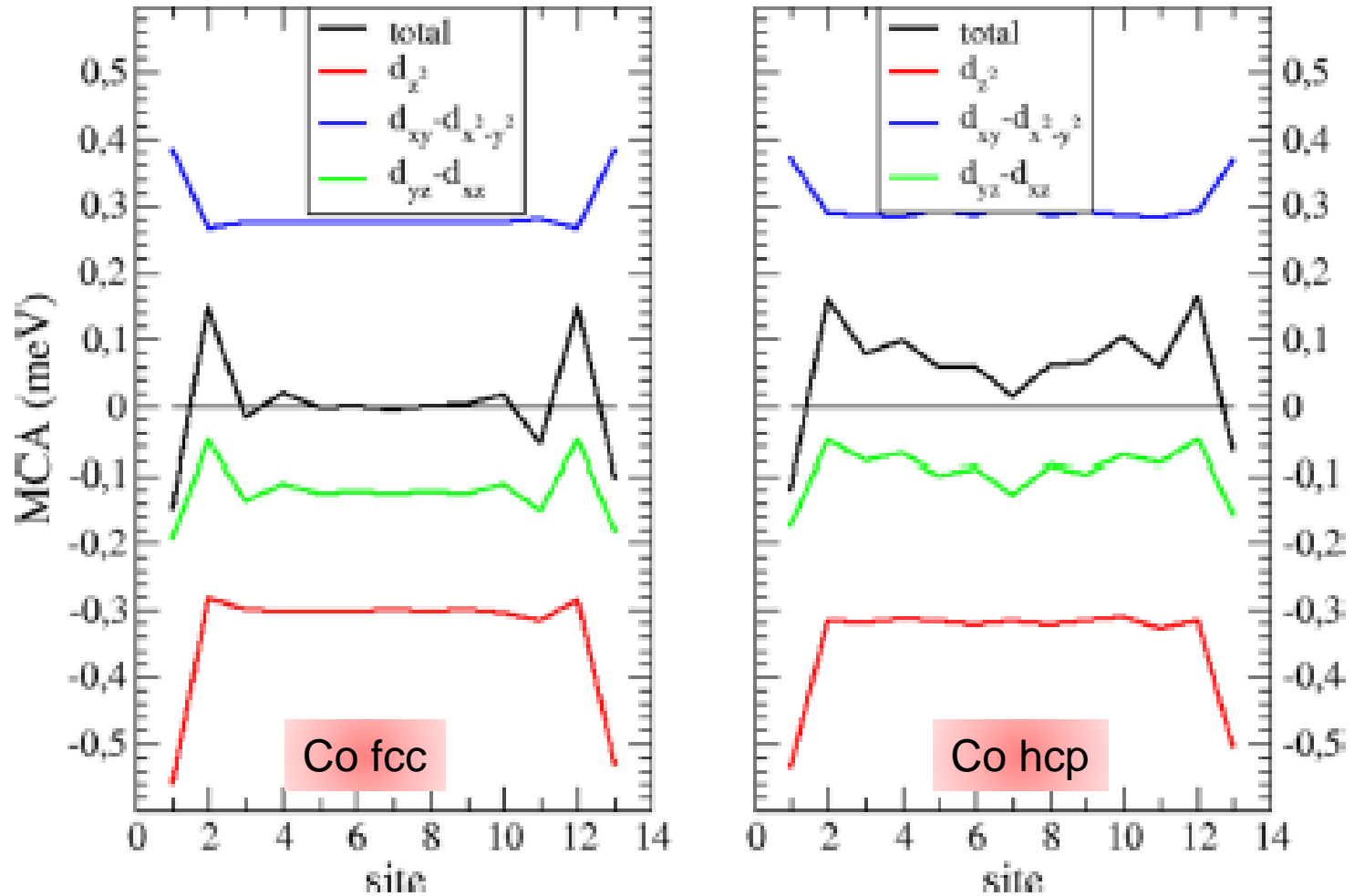
Co/Pt(111)



3 possible impact of Co thin film structure

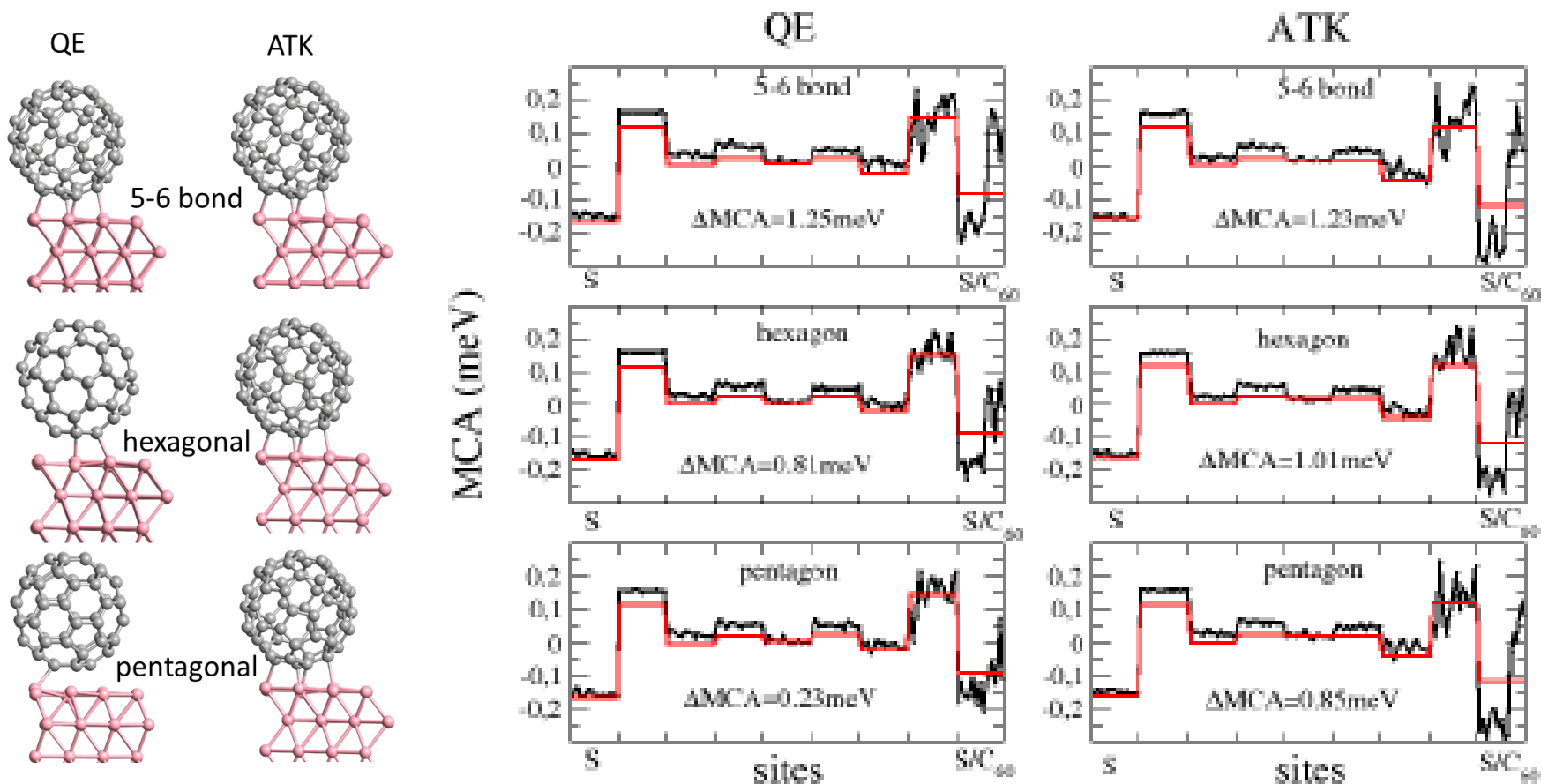
- Co Stacking: fcc vs hcp
 - fcc favored on Pt(111)
- $\neq C_{60}$ Adsorption geometry
 - \neq corrugation could induce
 - \neq adsorption site and geometry
- \neq Strain

FCC VS HCP



Rather modest effect

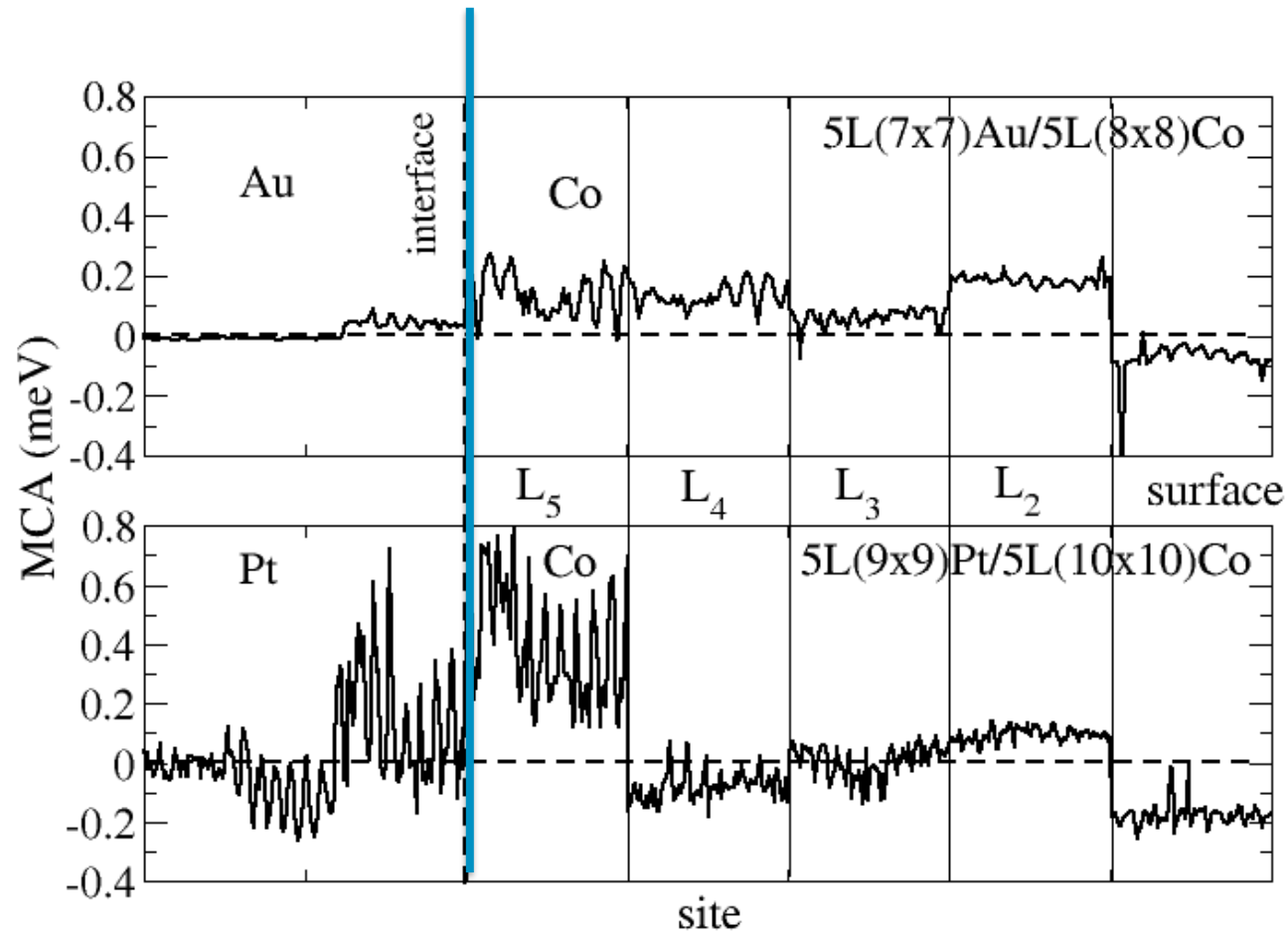
ADSORPTION GEOMETRY



Rather important influence but cannot be correlated to STM observations.

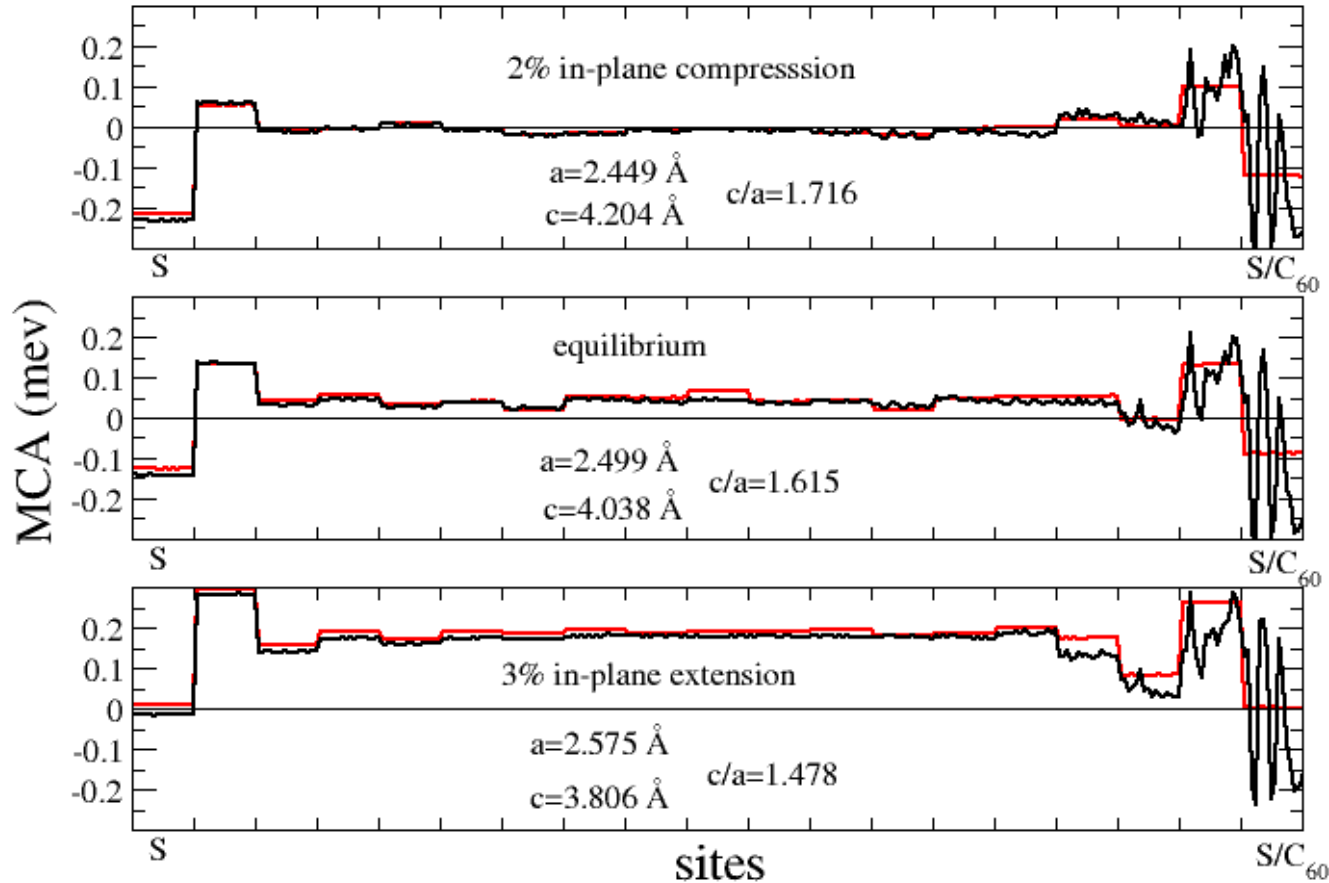
STRAIN

Co/Au(111) VERSUS Co/Pt(111)



strain induced by interface can affect the surface MCA

STRAIN C_{60}/Co



$$\Delta MCA > 1.5 \text{ meV}$$

Variation of MCA due to C_{60} adsorption is also very much influenced by strain

CONCLUSION & PERSPECTIVE



CONCLUSION & PERSPECTIVES

- Possible tuning of magnetic anisotropy
- Optimal choice?
 - Adequate combination molecule/substrate based on orbital engineering
 - Critical influence of strain → **straintronics!**
 - Optimal substrate: thin film thickness..
- Reversible tuning?
 - Adequate switchable molecule?

ACKNOWLEDGEMENTS



ANR, SPIROU

SPIn Resistance **On Ultimate** molecular devices



FET open Grant No. 766726



COncepts and **tools** in **Molecular spintronICS**

THANK YOU FOR YOUR ATTENTION

QUESTIONS?

COMMENTS?

